

## REVIEW



Cite this: *Phys. Chem. Chem. Phys.*,  
2023, 25, 30785

Received 29th August 2023,  
Accepted 23rd October 2023

DOI: 10.1039/d3cp04170k

rsc.li/pccp

# Recent advances in copper chalcogenides for CO<sub>2</sub> electroreduction

Wenjian Hu, <sup>ab</sup> Didier Grandjean, <sup>b</sup> Jan Vaes, <sup>ac</sup> Deepak Pant <sup>\*ad</sup> and  
Ewald Janssens <sup>\*b</sup>

Transforming CO<sub>2</sub> through electrochemical methods into useful chemicals and energy sources may contribute to solutions for global energy and ecological challenges. Copper chalcogenides exhibit unique properties that make them potential catalysts for CO<sub>2</sub> electroreduction. In this review, we provide an overview and comment on the latest advances made in the synthesis, characterization, and performance of copper chalcogenide materials for CO<sub>2</sub> electroreduction, focusing on the work of the last five years. Strategies to boost their performance can be classified in three groups: (1) structural and compositional tuning, (2) leveraging on heterostructures and hybrid materials, and (3) optimizing size and morphology. Despite overall progress, concerns about selectivity and stability persist and require further investigation. This review outlines future directions for developing the next-generation of copper chalcogenide materials, emphasizing on rational design and advanced characterization techniques for efficient and selective CO<sub>2</sub> electroreduction.

## 1. Introduction

The worldwide need for energy continues to grow while the alarming rise in greenhouse gas (GHG) emissions demand for sustainable and eco-friendly solutions.<sup>1–8</sup> Carbon dioxide (CO<sub>2</sub>) is a major GHG, contributing significantly to global warming and shifts in climate patterns. Creating practical and affordable solutions to capture and transform CO<sub>2</sub> into valuable chemicals and fuels is thus of urgent importance.<sup>9–11</sup> Among various CO<sub>2</sub> conversion methods that are being developed, electrochemical reduction has emerged as one of the most appealing approaches, given its notable energy efficiency, its potential for renewable energy integration, and its capacity to generate a variety of valuable products.<sup>12–27</sup> Specifically, the process of CO<sub>2</sub> electroreduction using Cu-based catalysts<sup>22,27–44</sup> has garnered tremendous attention because Cu demonstrates notable activity for C–C coupling, essential to the formation of multicarbon products.

Copper chalcogenides are a family of compounds that combine copper with elements from the chalcogen group, comprising sulfur, selenium, and tellurium. These materials have attracted considerable interest for multiple uses in semiconductors,

photoconductors, thermoelectric materials, and solar cells.<sup>45–50</sup> Copper chalcogenide bonds are highly covalent, although the chalcogenide is treated as a divalent anion (S<sup>2–</sup>, Se<sup>2–</sup>, or Te<sup>2–</sup>) in some cases. In a basic lattice cell, chalcogen atoms constitute the frame of a regular lattice, whereas the copper atoms are disordered, moving like a liquid and taking random sites in the chalcogen lattice.<sup>51,52</sup> This phenomenon leads to variations in stoichiometry, which is represented by the characteristic formula Cu<sub>2–x</sub>E (E = S, Se, Te) with possible *x* values ranging from 0 to 1.<sup>53,54</sup> The copper chalcogenide family encompasses various binary, ternary, and quaternary compounds, whose stoichiometries determine the crystal structure. The binary copper sulfide system has monoclinic (Cu<sub>31</sub>S<sub>16</sub> and Cu<sub>1.94–1.97</sub>S), cubic (Cu<sub>9</sub>S<sub>5</sub> and Cu<sub>1.8</sub>S), orthorhombic (Cu<sub>7</sub>S<sub>4</sub> and Cu<sub>1.75</sub>S), triclinic (Cu<sub>58</sub>S<sub>32</sub> and Cu<sub>1.81</sub>S), and hexagonal (CuS) crystal phases. The binary copper selenide system has cubic (Cu<sub>2</sub>Se and Cu<sub>1.8</sub>Se), tetragonal (Cu<sub>3</sub>Se<sub>2</sub>), hexagonal (CuSe and Cu<sub>0.87</sub>Se), and orthorhombic (Cu<sub>5</sub>Se<sub>4</sub> and CuSe<sub>2</sub>) phases. Known crystal phases of copper tellurides are hexagonal (Cu<sub>2</sub>Te), tetragonal (Cu<sub>7</sub>Te<sub>4</sub>, Cu<sub>4</sub>Te<sub>3</sub> and Cu<sub>3</sub>Te<sub>2</sub>), orthorhombic (CuTe), and pyrite-type (CuTe<sub>2</sub>).<sup>55–57</sup> In the context of the electrochemical CO<sub>2</sub> reduction, copper chalcogenides have shown an exceptional selectivity as a result of their distinct electronic and structural properties.

The synergistic effect between copper and the chalcogen elements is responsible for the remarkable electrocatalytic CO<sub>2</sub> reduction performance of these compounds. The presence of chalcogen atoms modulates their electronic structure, in particular the location of the Cu d-band center altering the adsorption behavior of the reaction intermediates, and subsequently influencing the catalyst's specificity and effectiveness.<sup>54,58</sup> The tunability

<sup>a</sup> Separation and Conversion Technology, Flemish Institute for Technological Research (VITO), Boeretang 200, 2400 Mol, Belgium. E-mail: deepak.pant@vito.be

<sup>b</sup> Quantum Solid-State Physics, Department of Physics and Astronomy, KU Leuven, Celestijnenlaan 200 D, 3001 Leuven, Belgium.  
E-mail: ewald.janssens@kuleuven.be

<sup>c</sup> Department of Solid-state Sciences, Ghent University, Krijgslaan 281/S1, 9000 Gent, Belgium

<sup>d</sup> Center for Advanced Process Technology for Urban Resource Recovery (CAPTURE), Frieda Saeyssstraat 1, 9052 Zwijnaarde, Belgium

of the copper chalcogenide properties enables the optimization of their electrocatalytic performance through various strategies, such as controlling their composition, adjusting their crystal structure and morphology, and creating heterostructures or hybrid materials. These approaches can be employed to customize the electronic configuration and catalytic performance of copper chalcogenides, enhancing their performance in CO<sub>2</sub> electroreduction by achieving higher selectivity, improved stability, and increased energy efficiency.<sup>59</sup>

The objective of this review is to examine the latest developments in copper chalcogenide catalysts for CO<sub>2</sub> electroreduction, focusing on the most significant contributions from the past five years. We discuss the latest approaches to enhance their efficacy as electrocatalysts, including structural and compositional tuning, exploiting heterostructures and hybrid materials, and optimizing their shape and particle size by synthesis methods.

Although significant advancements have been made in recent times, obstacles persist in increasing the preference for particular products and boosting the enduring stability of copper chalcogenide catalysts. Therefore, we also outline future perspectives on the development of next-generation copper chalcogenide materials, emphasizing the importance of rational design strategies and the integration of advanced characterization techniques to achieve efficient and selective CO<sub>2</sub> electroreduction.

## 2. Synthesis strategies for copper chalcogenides

Copper chalcogenides have been studied for several decades, initially for their applications in superconductors and thermoelectric materials. Their synthesis has evolved over time with

enhanced control over their composition, structure, and morphology. Solvothermal methods, vital for nanoparticle synthesis, allow the production of highly crystalline materials under controlled temperatures and pressures, with the ability to tailor compositions and structures. Electrosynthesis, offers fine control over the material properties and allows for *in situ* formation of compounds like copper chalcogenides on conductive substrates, but it does present challenges in scaling up. Meanwhile, cation exchange methods permit careful post-synthesis modifications, control over composition, and the synthesis of complex heterostructures and hybrid materials. By combining these techniques, researchers could overcome their individual limitations; solvothermal synthesis forms the initial nanostructures, electrosynthesis deposits additional material, and cation exchange refines their final properties, broadening the horizons for nanomaterial design and innovation (Fig. 1). Other methods, such as wet chemical method<sup>60–62</sup> pyrolysis<sup>63</sup> and heat treatment<sup>64,65</sup> have also been explored to produce copper chalcogenide materials with various catalytic properties for the electrochemical CO<sub>2</sub> reduction reaction (eCO<sub>2</sub>RR). Inspired by the water evaporation technique used in ref. 66, we could synthesize a variety of Cu<sub>2–x</sub>Se nanostructures, including nanosheets, nanoflowers, and nanowires. This was achieved by variations of the Cu/Se ratio of 3.75/19, 5/19 and 7.5/19, respectively (Fig. 2).

### 2.1 Solvothermal synthesis

Solvothermal synthesis is a prominent method for preparing copper chalcogenides for eCO<sub>2</sub>RR. It uses elevated temperatures and pressures in a sealed reactor to produce highly crystalline materials with tailored properties. By adjusting the

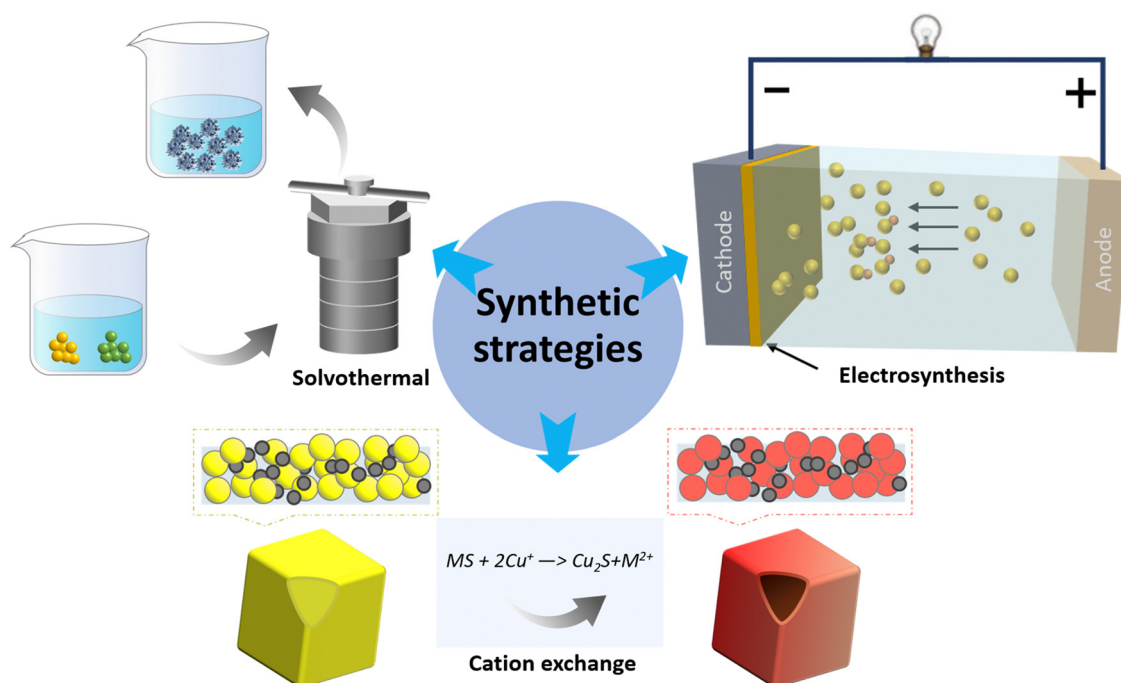


Fig. 1 Schematic diagram of synthesis strategies, including solvothermal, electrosynthesis and cation exchange. M stands for metal.

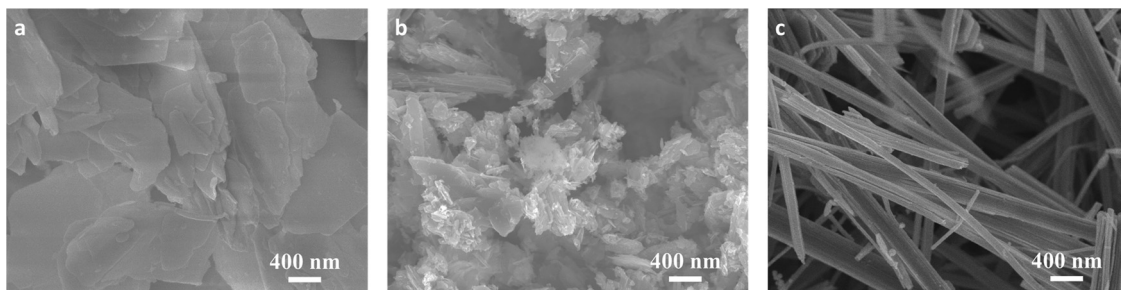


Fig. 2 Scanning electron microscopy images illustrating the variety of  $\text{Cu}_{2-x}\text{Se}$  nanostructure that can be grown by changing the Cu/Se ratio. (a) Nanosheets, (b) nanoflowers, and (c) nanowires synthesized from a Cu/Se precursor ratio of 3.75/19, 5/19 and 7.5/19, respectively.

reaction parameters, various nanostructures, such as nanoparticles, nanowires, and nanosheets could be obtained.

Solvothermal methods have extensively been used to synthesize copper sulfide nanostructures.<sup>63,67–82</sup> For example, Shinagawa *et al.*<sup>67</sup> reported that CuS nanocatalysts can be produced by *in situ* reductive reformation of CuS precursor materials. They developed a new method to create carbon-supported copper sulfide nanoparticles by mixing copper nitrate, thiourea, urea, and carbon black in water, and heating the mixture to precipitate the nanoparticles. Wang *et al.*<sup>73</sup> utilized a combination of solvothermal and electrochemical techniques to create  $\text{Cu}_2\text{SnS}_3$  nanosheets. This involved dissolving  $\text{SnCl}_4 \cdot 5\text{H}_2\text{O}$  and  $\text{Cu}(\text{CH}_3\text{COO})_2 \cdot \text{H}_2\text{O}$  within a combined concoction of ethylene glycol and water, agitating the solution, adding thioacetamide, and heating the mixture in a Teflon-lined autoclave. The synthesis of CuS nanosheets followed a similar procedure, while  $\text{SnS}_2$  nanosheets were prepared using  $\text{SnCl}_4 \cdot 5\text{H}_2\text{O}$ , sodium dodecyl benzenesulfonate, and L-cysteine. Following  $\text{eCO}_2\text{RR}$ , the samples experienced self-adapted phase separation, resulting in the formation of  $\text{SnO}_2@\text{CuS}$  and  $\text{SnO}_2@\text{Cu}_2\text{O}$  heterojunctions. Zhang *et al.*<sup>69</sup> created CuS nanosheets by mixing 6 mL of 0.5 M thioacetamide with 25 mL of 0.2 M  $\text{Cu}(\text{NO}_3)_2$  ethanol solutions. The mixture was heated to 90 °C for 3 hours, whereafter the resulting black precipitate was collected. CuS hollow particles were synthesized using a modified method in which thioacetamide dissolved in ethanol is added to a dispersion of MOF-199 in ethanol that is subsequently heated at 90 °C for 1 hour. TEM and high-angle annular dark-field scanning TEM (HAADF-STEM) images showed that CuS hollow particles

have an average size of 1.3  $\mu\text{m}$  and are composed of CuS nanosheets with a thickness of about 3 nm.

The hydrothermal synthesis was also used to produce copper selenide nanomaterials.<sup>83–89</sup> Yang *et al.*<sup>83</sup> combined  $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$ ,  $\text{Na}_2\text{SeO}_3$ , and hydroxylamine in a diethylenetriamine and deionized water mixture with varying volume ratios to synthesize  $\text{Cu}_{2-x}\text{Se}(y)$  nanocatalysts. After stirring for 30 minutes, the mixture was warmed in a Teflon-coated pressure vessel at 180 °C during 15 hours and the resulting products were washed, and dried. The obtained  $\text{Cu}_{1.63}\text{Se}$  nanoparticles had a typical size of 50 nm (see Fig. 3). Wang *et al.*<sup>85</sup> prepared  $\text{CuInSe}_2$  using a modified solvothermal method. Oleylamine was degassed and purged with  $\text{N}_2$  before combining  $\text{CuCl}$ ,  $\text{InCl}_3$  and Se in a flask. The mixture was heated and stirred before the products were washed, and dried. Duan *et al.*<sup>88</sup> synthesized  $\text{Cu}_2\text{Se}$  nanosheets using a hydrothermal treatment.  $\text{BiCuSeO}$  nanosheets were produced by mixing  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  with  $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$ , KOH and NaOH, heating the mixture in a Teflon-lined vessel, and washing and drying the obtained precipitate. The average thickness of the  $\text{BiCuSeO}$  sheets was 9.2 nm. High-resolution TEM images display two perpendicular sets of lattice fringes with a spacing of 0.28 nm, verifying the successful synthesis of ultrathin  $\text{BiCuSeO}$  single-crystal nanosheets. Ding *et al.*<sup>87</sup> synthesized K-doped  $\text{Cu}_2\text{Se}$  nanosheet arrays on a Cu foam. The foam was cleaned, soaked in a mixture of Se powder,  $\text{NaBH}_4$ , NaOH, deionized water, and KBr, and heated in an autoclave. Various KBr amounts were used to produce different potassium dopant contents, while undoped pure  $\text{Cu}_2\text{Se}$  was obtained without KBr. 11.2% K-doped  $\text{Cu}_2\text{Se}$  had

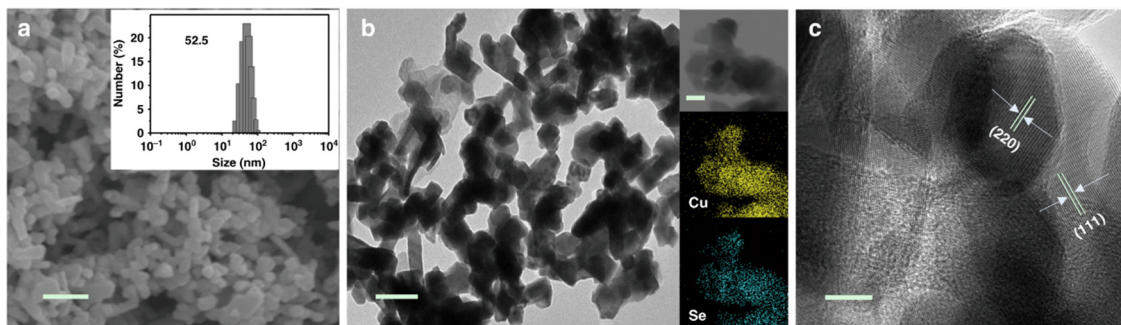


Fig. 3 Structural characterization of  $\text{Cu}_{1.63}\text{Se}$  nanocatalysts. (a) SEM image with a 200 nm scale bar. The inset presents the size distribution as obtained by dynamic light scattering (DLS) measurements. (b) TEM image with a 100 nm scale bar, including elemental mappings as insets. (c) HRTEM image with a 10 nm scale bar. Copyright 2019 Nature Publishing Group.<sup>83</sup>

a honeycomb-like structure and a thickness of 1.4  $\mu\text{m}$ , while pure  $\text{Cu}_2\text{Se}$  had a thickness of 1.8  $\mu\text{m}$ . Those nanosheets display a plane spacing of 0.205 nm, which correlates with the (220) facet of cubic  $\text{Cu}_2\text{Se}$ .

The latter method is widely used for the synthesis of various types of materials including nanoparticles, single crystals, and complex structures. It provides the advantage of synthesizing high-quality materials at relatively lower temperatures than conventional methods. However, the process is often time-consuming and requires careful handling due to high pressures involved. In addition, the solvents used can be harmful, requiring careful management.

## 2.2 Electrosynthesis

*In situ* electrochemical synthesis, including electrochemical deposition, recently gained traction for the synthesis of copper chalcogenide nanomaterial.<sup>75,81,90–94</sup> Redox reactions cause selective ion deposition on either the cathode or the anode surface. Key benefits of electrochemical deposition include simplicity of the operating procedure, precise control over particle size, and environment friendly outcomes without corresponding oxidation or reduction byproducts. As the understanding of catalyst synthesis methods evolves, many catalysts are not produced using a single method but by combining solvothermal and *in situ* electrochemical synthesis techniques.

Studies emphasize the importance of the catalysts' elemental composition and their intricate microstructures. Cheng *et al.*<sup>81</sup> demonstrated that introducing surface nitrogen can notably improve the electrocatalyst's performance by raising the concentration of active sites and optimizing their electronic structure. The obtained N-enriched Sn/SnS nanosheets demonstrated a five-fold rise in current density and a 2.45-fold increase in faradaic efficiency (FE) compared to their unmodified SnS<sub>2</sub>-derived counterparts. The formate production rate was 14 times higher in the N-enriched sheets than in the unmodified ones, reaching 1358  $\mu\text{mol h}^{-1} \text{cm}^{-2}$ . The surface nitrogen-injection technique was also effectively employed with other precursors such as CuS and In<sub>2</sub>S<sub>3</sub>, demonstrating its potential as a broader approach. Peng *et al.*<sup>75</sup> synthesized double sulfur vacancy-rich  $\text{CuS}_x$  nanosheets for the electro-conversion of  $\text{CO}_2$  into C3 species. In a first stage CuS nanosheets were prepared by dissolving  $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$  and thioacetamide in separate solutions that were then combined and subsequently heated and washed. Sulfur vacancies were then introduced in the CuS powder by mixing it with polyvinylidene fluoride and coating it on a copper foil that was used as a cathode in a lithium-ion battery. A 0.044  $\text{mA cm}^{-2}$  current density was applied within a 0.01–3 V *vs.*  $\text{Li}^+/\text{Li}$  voltage range.  $\text{CuS}_x$  catalysts with various sulfur vacancy densities were obtained after 1, 10, and 100 charge–discharge cycles. HAADF-STEM images revealed that the  $\text{CuS}_x$ -10-cycle nanosheets display a hexagonal phase where the 0.283 nm spacing corresponds to CuS (103) facets. The (103) planes were also seen on  $\text{CuS}_x$ -1-cycle and  $\text{CuS}_x$ -100-cycle edges. Deng *et al.*<sup>90</sup> developed a method for the electrochemical deposition of  $\text{CuS}_x$  films on polished Cu disks by utilizing square wave voltammetry.

The synthesis procedure involved 2000 cycles with a plating solution containing a mixture of  $\text{CuSO}_4 \cdot 5\text{H}_2\text{O}$ , thiourea, and HCl. Active  $\text{CuS}_x$  and desulfurized  $\text{CuS}_x$  were prepared for electrochemical  $\text{CO}_2$  reduction to investigate the role of sulfur. Phillips *et al.*<sup>91</sup> prepared sulfide-derived-Cu working electrodes *via* pulsed electrodeposition. The active sulfide-derived-Cu layer was formed *in situ* during the first  $\sim 15$  minutes of electrolysis. Guo *et al.*<sup>93</sup> synthesized  $\text{Cu}_2\text{S}$  hollow nanocubes with a significant presence of sulfur vacancies by electrochemical reduction (see Fig. 4).  $\text{Cu}_2\text{S}_{1-x}$  hollow nanocubes were obtained by electro-reducing  $\text{Cu}_2\text{S}$  hollow nanocubes loaded on a gas diffusion electrode (GDE) at  $-0.3$  V *vs.* RHE for 30 minutes. Li *et al.*<sup>94</sup> prepared Ag and S co-doped  $\text{Cu}_2\text{O}/\text{Cu}$  (Ag, S- $\text{Cu}_2\text{O}/\text{Cu}$ ) using an electrochemical reduction method at room temperature in a conventional H-cell. Ag- $\text{Cu}_2\text{S}$  was first prepared through a cation exchange method and used as the working electrode. Electrochemical reduction was then carried out by applying a potential of  $-1.6$  V *versus* a saturated calomel electrode for 30 minutes, while bubbling  $\text{CO}_2$  and stirring continuously.  $\text{Cu}_2\text{O}/\text{Cu}$ , S- $\text{Cu}_2\text{O}/\text{Cu}$ , and Ag- $\text{Cu}_2\text{O}/\text{Cu}$  were fabricated using the same procedure, with  $\text{Cu}_2\text{O}$ ,  $\text{Cu}_2\text{S}$ , and Ag- $\text{Cu}_2\text{O}$  as working electrodes, respectively.

Electrosynthesis can be more environment friendly than conventional methods since in many cases the use of electricity does not require potentially harmful chemical reagents. On the downside, it can require extensive optimization of individual processes and, for certain reactions, achieving a high selectivity can be challenging.

## 2.3 Cation exchange

Cation exchange, a wet chemical method,<sup>61,62,95–97</sup> is a relatively new approach for synthesizing copper sulfide nanocatalysts for  $\text{eCO}_2\text{RR}$ . By selectively replacing the cations of a host material with copper ions, CuS nanosheets and nanoparticles with controlled properties can be produced.

He *et al.*<sup>96</sup> introduced electrochemically-driven cation exchange (ED-CE), which enables converting a prearranged  $\text{CoS}_2$  template into  $\text{Cu}_2\text{S}$  through a three-step process using a standard three-electrode system. A metal–organic framework (MOF) consisting of a Co-containing zeolitic imidazolate framework (Co-ZIF-L) grown on a carbon cloth and Cu foil is electrochemically converted into cobalt sulfides and finally cations are exchanged. In this configuration, the cobalt sulfide film functioned as working electrode and cation exchange template, while a platinum foil and a silver/silver chloride electrode saturated with potassium chloride served as counter and reference electrodes, respectively. To replace the cobalt cations of  $\text{CoS}_2$  with copper ions, a 40 mL dimethylformamide solution containing 3 mM copper nitrate trihydrate and 0.1 M lithium perchlorate was used as electrolyte. Chronoamperometry was employed to drive the cation exchange processes, which was performed first at  $-0.2$  V (*vs.* Ag/AgCl) and thereafter at  $-0.6$  V (*vs.* Ag/AgCl) until a targeted value of electrical charge per electrode area was reached. The synthesis procedure is schematically depicted in Fig. 5. This method ensures that the synthesized  $\text{Cu}_2\text{S}$  catalyst keeps the same structure as the  $\text{CoS}_2$  template, while maintaining a high density of grain



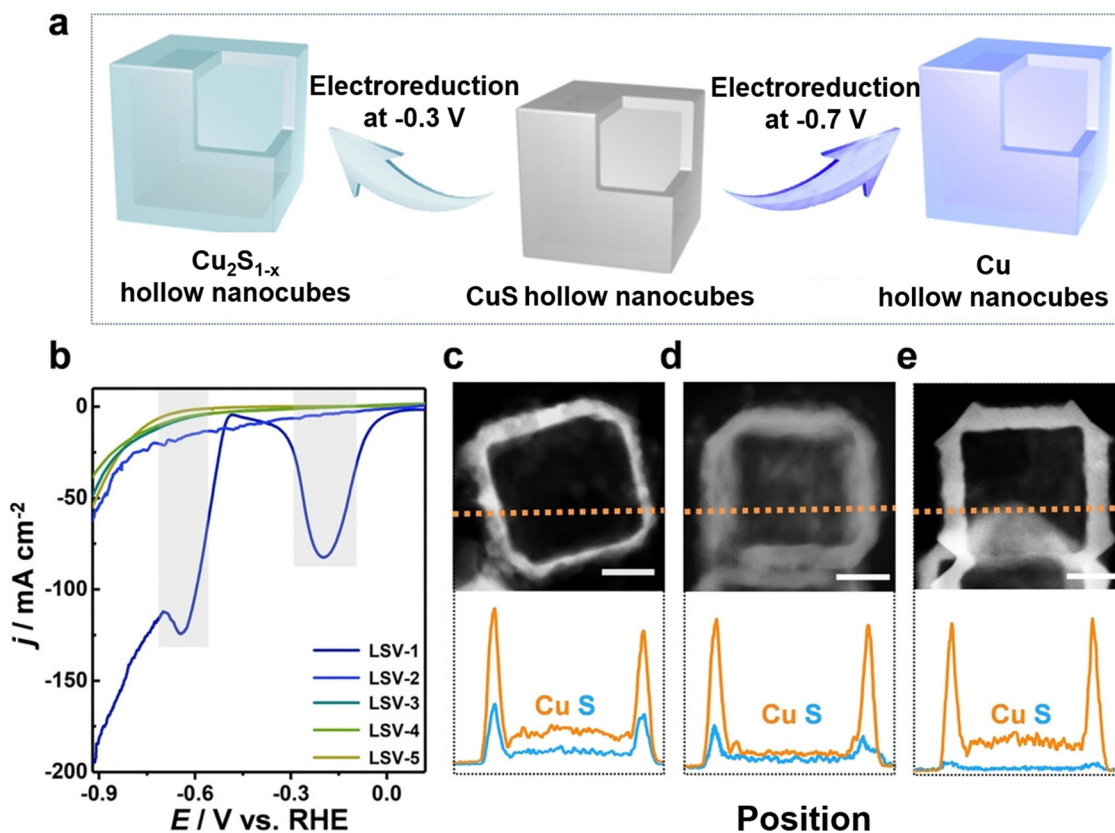


Fig. 4 (a) Schematic representation of the electrochemical reduction process of CuS hollow nanocubes into their Cu<sub>2</sub>S<sub>1-x</sub> and Cu counterparts. (b) Subsequent LSV curves for the CuS hollow nanocubes (numbered LSV-1 to LSV-5). TEM images along with corresponding EDX line-scanning profiles for (c) CuS, (d) Cu<sub>2</sub>S<sub>1-x</sub>, and (e) Cu hollow nanocubes, all featuring a 30 nm scale bar. Figure adapted from ref. 93. Copyright 2022 Wiley VCH.

boundaries. Li *et al.*<sup>97</sup> developed a simple cation exchange technique to create Ag/Cu electrocatalysts using copper sulfide nanosheets as growth template. The crystal structure of Cu<sub>2-x</sub>S nanosheets with lateral dimensions of 100 nm and a 14 nm thickness gradually transformed from Cu<sub>7</sub>S<sub>4</sub> to Ag<sub>2</sub>S, with an Ag/Cu mass ratio varying between 0.3 and 25 upon increasing the Ag<sup>+</sup> concentration in the exchange solution. Following the cation exchange process, the nanosheets retain their original morphology, but exhibit shape distortion with increasing Ag content.

We conclude that cation exchange is an attractive method to modify the chemical composition and hence the properties of the material without disrupting its overall structure. It has potential to fine-tune materials for specific applications. However, the method's success highly depends on the cation exchange kinetics and thermodynamics, which can be complex and hard to control.

### 3. Recent strategies to enhance the performance of copper sulfide in eCO<sub>2</sub>RR

Copper sulfide has emerged as an effective catalyst for the eCO<sub>2</sub>RR, but further improvement is needed. Notably, the

hydrogen evolution reaction (HER) is a competitive reaction to eCO<sub>2</sub>RR that hinders its efficiency. Research has shown a linear correlation between the binding energies of COOH\*/HCOO\* and H\* on most eCO<sub>2</sub>RR catalysts, negatively impacting the selectivity.<sup>98–102</sup> Breaking this correlation is imperative for improving copper sulfide's effectiveness as an eCO<sub>2</sub>RR catalyst for industrial applications. Recent strategies to enhance the selectivity and activity of CuS include (i) structural and compositional tuning through sulfur vacancy engineering and foreign metal ions doping, (ii) combining CuS with other materials in heterostructures or hybrid materials to introduce new active sites and improve the catalytic performance through synergistic effects, and (iii) size and morphology control, for example to create nanosheets or hollow structures with optimized surface area and enhanced accessibility of the active sites. These approaches offer a guide for developing effective CuS catalysts for eCO<sub>2</sub>RR, advancing eco-friendly conversion methods. A summary of cutting-edge Cu sulfide catalysts, including information on their preparation method, operating conditions and product preferences, is provided in Table 1.

#### 3.1 Structural and compositional tuning

To enhance copper sulfide's eCO<sub>2</sub>RR catalytic performance, various structural and compositional tuning

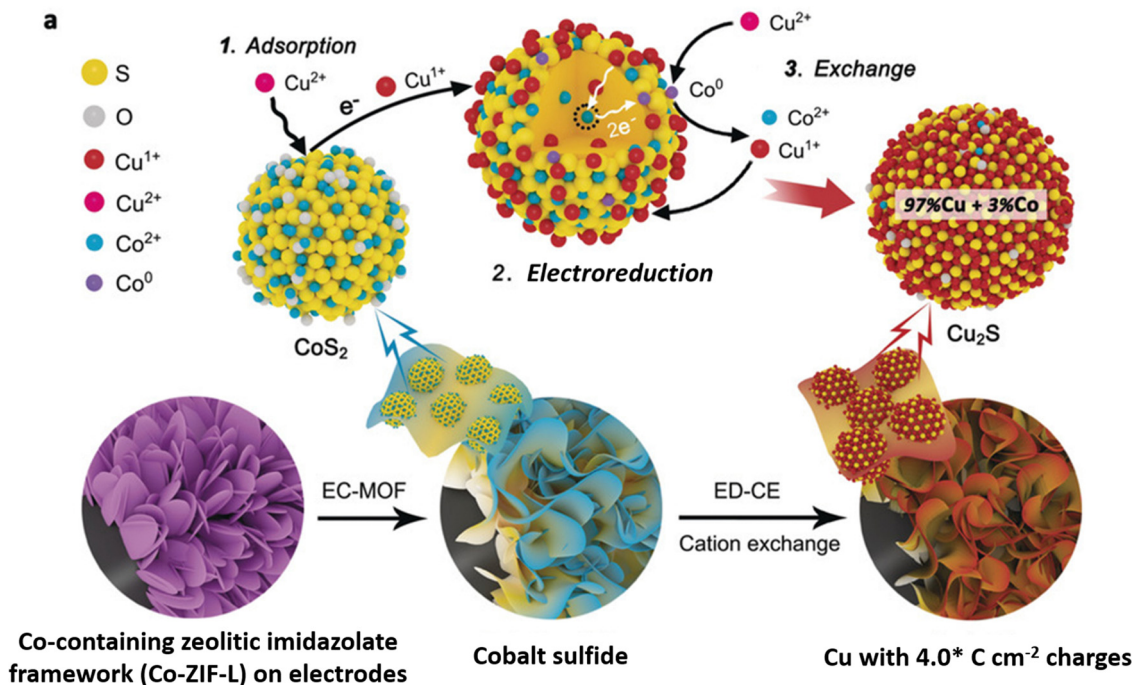


Fig. 5 Experimental routes and the mechanisms involved in cation exchange driven by electrochemical means. Figure adapted from ref. 96. Copyright 2020 Wiley-VCH.

strategies,<sup>61,65,67,69,75,79,81,82,90,91,93,94,96,97,103</sup> can be employed to adjust its electronic structure and morphology.

We outline two key strategies. The first one is by creating sulfur vacancies in the material during or after its synthesis by controlling the preparation conditions or through post-synthesis treatments, respectively. Their presence increases the likelihood of CO\* formation on the catalyst surface. The second is by doping CuS with metal cations. Ag doping, for instance, induces synergistic interactions between Ag<sub>2</sub>S and CuS that suppress HER and improve the selectivity towards CO<sub>2</sub> reduction. Metal doping can be achieved by co-precipitation, ion exchange, or electrochemical deposition.

Cheng *et al.*<sup>81</sup> found that surface nitrogen-injection can increase the number of active sites and optimize the material's electronic structure. N-Enriched Sn/SnS nanosheets demonstrated significant improvements in current density and FE compared to their pristine versions. The improved catalytic activity was confirmed through density functional theory (DFT) calculations to be a result of the alteration in the valence of the Sn sites. The approach is applicable to other metal sulfide precursors as well. Peng *et al.*<sup>75</sup> showed that double sulfur vacancies (DSVs) in the CuS(100) planes could serve as active electrocatalytic centers for CO<sub>2</sub>RR by favoring the creation of *n*-propanol's key \*C3 intermediate. They applied a lithium electrochemical tuning method, using the CuS as the cathode and a lithium foil as the anode, yielding promising results for the controlled creation of ion vacancies (see Fig. 6). DFT calculations suggest that the creation processes of \*HCOOH and CO\* are in competition with the presence of sulfur vacancies, increasing the formation probability of the latter one. During continuous electrochemical CO<sub>2</sub>RR testing, the CuS<sub>x</sub>-DSV catalyst maintained a

FE<sub>*n*-PROH</sub> of 12.1% for 10 hours at −1.05 V vs. RHE, corresponding to a retention of approximately 78%. Meanwhile, an increase in the hydrogen production activity was observed, indicated by a continuous rise in the current. Deng *et al.*<sup>90</sup> found that sulfur plays a crucial role in this process, with active CuS<sub>x</sub> exhibiting high selectivity and activity towards formate (FE HCOO<sup>−</sup> = 75% and *j* HCOO<sup>−</sup> = −9.0 mA cm<sup>−2</sup> at −0.9 V). Operando Raman spectroscopy revealed that sulfur dopants suppress the formation of CO intermediate during eCO<sub>2</sub>RR, resulting in a decreased FE for CO, hydrocarbons, and alcohols. Supporting DFT calculations indicated that the presence of sulfur on the copper surface modified the adsorption strengths of HCOO\*, promoting formate production while suppressing that of CO. Phillips *et al.*<sup>91</sup> suggested that the selective formation of formate is due to the stronger binding of the CO intermediate originating from the sulfur atoms. Guo *et al.*<sup>93</sup> reported a Cu<sub>2</sub>S<sub>1−*x*</sub> catalyst, featuring numerous Cu<sup>δ+</sup> species that can produce ethanol at ultralow overpotential with an outstanding selectivity and FE with a 6-hour stability. The integrated findings from *in situ* attenuated total reflection Fourier transformed infrared spectroscopy and DFT calculations demonstrate that the robust electron-donating ability of Cu<sup>δ+</sup> sites contributes to reducing the Gibbs free energy for the C–C coupling step involved in generating ethanol, while concurrently enhancing that associated with formate formation.

Li *et al.*<sup>94</sup> found that the synergistic interactions between S<sup>2−</sup> anions and Ag<sup>+</sup> cations dopants and the Cu<sub>2</sub>O/Cu host material, not only enhanced the Ag, S-Cu<sub>2</sub>O/Cu electrocatalyst selectivity and current density for methanol formation but also ensured its stability over 24 hours. More specifically, S anions were found to adjust the electronic structure and morphology of the material, while according to their DFT calculations Ag<sup>+</sup> cations

**Table 1** Performance summary of recently reported copper sulfide eCO<sub>2</sub>RR catalysts where partial current densities were not provided in the literature they were calculated by multiplying the total current density with the Faradaic efficiency of the main product

Preparation method	Catalysts	Electrolyte	<i>E</i> (V vs. RHE)	Partial current density (mA cm <sup>-2</sup> )	Product (FE)	Ref.
Solvothermal	Cu <sub>2</sub> O/CuS Nanocomposites	0.1 M KHCO <sub>3</sub>	−0.9	15.3	HCOO <sup>−</sup> (67.6%)	74
	N-Enriched CuS	0.1 M KHCO <sub>3</sub>	−0.8	NA	HCOO <sup>−</sup> (~88%)	81
	Copper sulfide nanocrystals	0.1 M KHCO <sub>3</sub>	−1	2.1	HCOO <sup>−</sup> (52.2%)	82
	CuS@SnO <sub>2</sub> nanosheets	0.5 M KHCO <sub>3</sub>	−1.0	~150	HCOO <sup>−</sup> (83.4%)	73
	CuS@MoS <sub>2</sub> heterostructures	CO <sub>2</sub> saturated electrolyte	−0.69	NA	NA	68
	CuS/N, S-rGO	0.5 M KHCO <sub>3</sub>	−0.63	~5	HCOO <sup>−</sup> (82%)	71
	CuS-thiourea	0.1 M KHCO <sub>3</sub>	−0.51	~0.8	CO (72.67%)	80
	Cu <sub>2</sub> S nanocrystal-decorated Cu nanosheets	0.1 M KHCO <sub>3</sub>	−1.2	~20.7	C <sub>2</sub> H <sub>5</sub> OH (46%)	78
	Sm <sub>x</sub> -CuS <sub>y</sub>	1 M KOH	−0.52	276	HCOO <sup>−</sup> (92.1%)	77
	Sulfur tailored Cu <sub>2</sub> O	0.1 M KHCO <sub>3</sub>	−1	16.3	HCOO <sup>−</sup> (67.2%)	70
Electrosynthesis	Active CuS <sub>x</sub>	0.1 M KHCO <sub>3</sub>	−0.9	9	HCOO <sup>−</sup> (75%)	90
	Sulfide-derived Cu	0.1 M KHCO <sub>3</sub>	−0.8	~2.5	HCOOH (~60%)	91
	Cu <sub>2</sub> S <sub>1−x</sub> with abundant Cu <sup>δ+</sup>	0.5 M KHCO <sub>3</sub>	−0.3	~2.5	C <sub>2</sub> H <sub>5</sub> OH (73.3%)	93
	Dual-doped Ag, S-Cu <sub>2</sub> O/Cu	BMIMBF <sub>4</sub> /H <sub>2</sub> O	−1.18	82.7	CH <sub>3</sub> OH (67.4%)	94
Cation exchange	CuS <sub>x</sub>	0.1 M KHCO <sub>3</sub>	−0.6	2.2	HCOO <sup>−</sup> (71.8%)	62
	3D Cu <sub>2</sub> S	0.1 M NaHCO <sub>3</sub>	−0.9	19.1	HCOO <sup>−</sup> (87.3%)	96
	Ag/Cu sulfide	0.05 M K <sub>2</sub> CO <sub>3</sub>	−1.2	5.1	C <sub>2</sub> H <sub>4</sub> (34%)	97
Wet chemistry	Cu and CuS nano-particles@graphene nanoflakes	0.1 M KHCO <sub>3</sub>	−0.6	2.3	HCOO <sup>−</sup> (42.5%)	79
	SnO <sub>2</sub> confined on CuS nanosheets	0.1 M KHCO <sub>3</sub>	−0.8	~17	syngas (>85%)	72
	Sulfide-derived CuCd	0.1 M KHCO <sub>3</sub>	−0.8	0.6	C <sub>2</sub> H <sub>5</sub> OH (~32%)	76
	Sulfur vacancy-rich CuS	0.1 M KHCO <sub>3</sub>	−1.05	~3.3	<i>n</i> -Propanol (15.4%)	75
	Hierarchical CuS hollow polyhedrons	0.5 M K <sub>2</sub> SO <sub>4</sub>	−0.6	~16	HCOO <sup>−</sup> (~50%)	69
	Sulfur-modified copper catalysts	0.1 M KHCO <sub>3</sub>	−0.8	~21	HCOO <sup>−</sup> (80%)	67
Other methods						
Thermal oxidation	CuS decorated CuO heterostructure	0.1 M KHCO <sub>3</sub>	−0.7	20	HCOO <sup>−</sup> (84%)	95
<i>In situ</i> crystallization	Cu <sub>1.81</sub> S particles on carbon nanotubes	0.5 M KHCO <sub>3</sub>	−0.67	3.1	HCOO <sup>−</sup> (82%)	60
Chemical bath deposition	CuS nanosheet arrays	0.5 M KHCO <sub>3</sub>	−0.7	~50	HCOO <sup>−</sup> (70.2%)	61
Impregnation and annealing	Cu <sub>2−x</sub> S derived nanoparticles@C	0.5 M KHCO <sub>3</sub> and/or 0.5 M KCl	−0.78	0.18	HCOO <sup>−</sup> (12%)	65

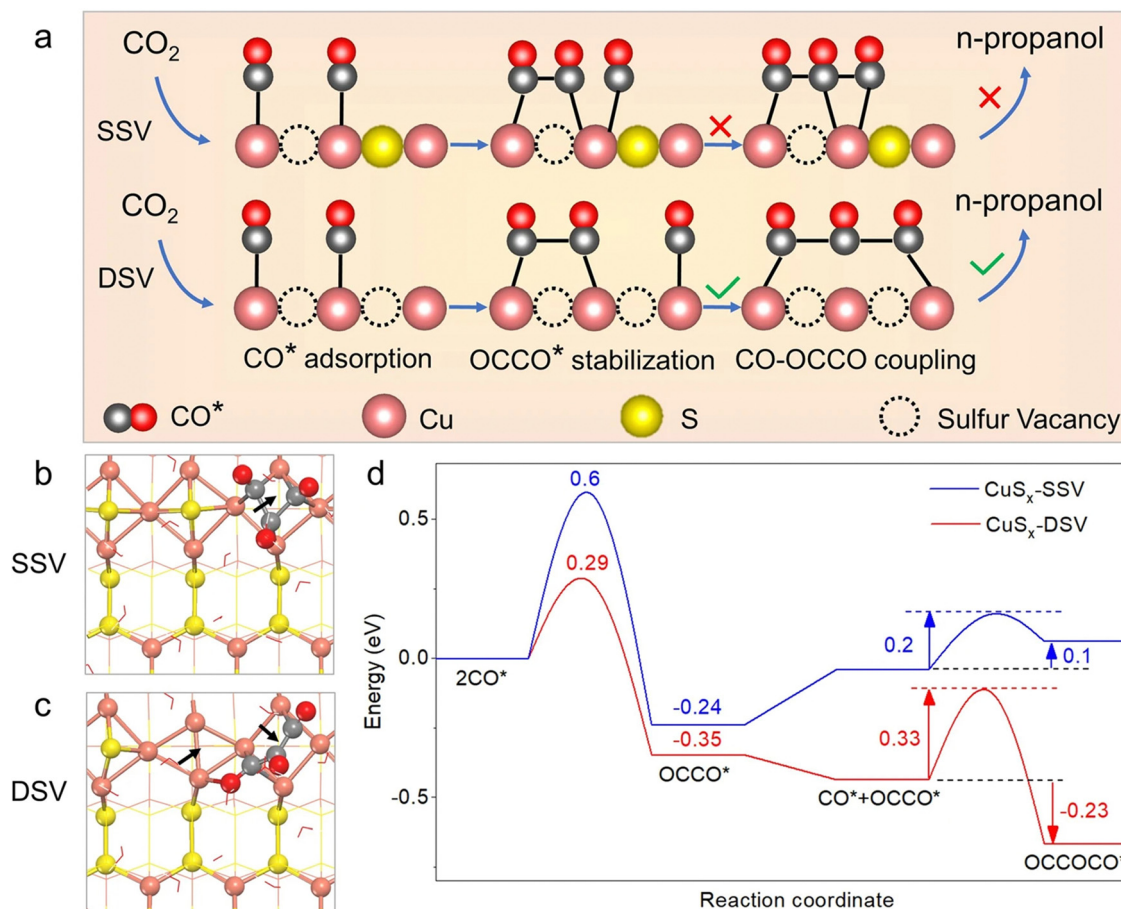
suppressed HER. Zhang *et al.*<sup>69</sup> studied the eCO<sub>2</sub>RR catalytic properties of CuS hollow polyhedra that experience structural transformation, as they evolve into a sulfur-doped metallic Cu catalyst. The *in situ* evolved catalyst exhibited a high durability and formate selectivity, maintaining stable operation for 36 hours at a formate partial current density of approximately 16 mA cm<sup>−2</sup> (at −0.6 V vs. RHE). This stability and selectivity were underscored by DFT calculations, which highlighted the significant role of sulfur-doped Cu (111) facets in its performance.

### 3.2 Synergistic effects of heterostructures and hybrid materials

Advancements in material science have demonstrated that heterostructures, comprising multilayer materials, and hybrid materials allow to optimize the catalytic CO<sub>2</sub>RR performance, selectivity, and stability through synergistic effects.<sup>68,71–74,76–78,95</sup> For example, CuS-based heterostructures with other semiconductors or metal catalysts can have optimized charge transfer, energy band alignment, and beneficial interfacial interactions to enhance their eCO<sub>2</sub>RR performance. The same applies to hybrid materials that combine CuS with conductive polymers, metal–organic frameworks, or graphene-based materials.

Wang *et al.*<sup>73</sup> discovered that Cu<sub>2</sub>SnS<sub>3</sub> nanosheets undergo a self-adaptive phase separation during the electrochemical process, resulting in the formation of stable SnO<sub>2</sub>@CuS and SnS<sub>2</sub>@Cu<sub>2</sub>O heterojunctions. Because Sn<sup>4+</sup> cations can be easily drawn by the negative potential, they migrate from the core to the surface of the nanosheet, replacing coordinated S atoms with O to form SnO<sub>2</sub> nanoparticles. Meanwhile, Cu<sup>+</sup> ions, carrying S atoms, concentrate in the center, forming CuS (Fig. 7). They identified that activated Sn is responsible for CO<sub>2</sub> reduction in CuS@SnO<sub>2</sub> ternary sulfides, which facilitates a self-adaptive transformation. By harnessing the synergistic effect of a stable heterojunction structure and electron self-flow, the optimized catalysts efficiently convert CO<sub>2</sub> to formate with 84.1% FE for 10 hours. This is supported by DFT calculations demonstrating that at the SnO<sub>2</sub> and CuS/Cu<sub>2</sub>O interface, delocalized Sn<sup>4+</sup> electrons transfer to Cu<sup>+</sup>, thereby enhancing the HCOO\* affinity, promoting H<sub>2</sub>O dissociation, and stabilizing Sn<sup>4+</sup> sites, which collectively improve the CO<sub>2</sub>RR activity and selectivity. This work revealed the active center of the ternary sulfide, offering insights for designing better electrocatalysts. Wang *et al.*<sup>74</sup> presented an innovative Cu<sub>2</sub>O/CuS composite catalyst that exhibits a 67.6% FE and a notable partial current density of 15.3 mA cm<sup>−2</sup> at −0.9 V *versus* RHE for formate.





**Fig. 6** (a) Schematic illustration of the conversion of CO<sub>2</sub> to n-propanol on CuS<sub>x</sub> with double sulfur vacancies (DSV) and related calculations. The process involves n-propanol formation on adjacent CuS<sub>x</sub>-DSV through CO–CO dimerization and CO–OCCO coupling. Structural diagrams displaying the optimized OCCOCO\* intermediate configurations on the (100) surface of CuS<sub>x</sub> with (b) a single sulfur vacancy (SSV) and (c) CuS<sub>x</sub>-DSV. Arrows indicate sulfur vacancy positions. (d) Energy diagrams for CuS<sub>x</sub>-SSV (blue curve) and CuS<sub>x</sub>-DSV (red curve) at 0 V vs. RHE. Copyright 2021 Nature Publishing Group.<sup>75</sup>

This catalyst maintains an average FE of 62.9% for at least 30 hours at the same potential, outperforming other Cu, CuS, and Cu<sub>2</sub>O catalysts in terms of selectivity and stability. The incorporation of CuS is believed to stabilize Cu<sub>2</sub>O during CO<sub>2</sub> reduction. DFT calculations imply that the CuS (110) surface favors the formation of formate over CO. In another study, Wang *et al.*<sup>72</sup> designed a novel hierarchical structure combining SnO<sub>2</sub> nanoparticles with CuS nanosheets, improving the electroreduction of CO<sub>2</sub> into a versatile syngas mixture (CO/H<sub>2</sub> ratio varying in the range of 0.11–3.86). The material achieved a remarkable FE > 85%, a significant turnover frequency of 96.1 h<sup>-1</sup>, and a durability of more than 24 hours. Experimental characterization and theoretical calculations demonstrated that various factors like size, morphology and the density of the SnO<sub>2</sub>–CuS interface contribute to the enhanced catalytic performance. In a study by Mosali *et al.*,<sup>76</sup> a near 30% FE for ethanol production for 6 hours was reported (at –0.8 V versus RHE), using the sulfide-derived-CuCd<sub>2</sub> catalyst in 0.1 M KHCO<sub>3</sub>. This notable ethanol selectivity was attributed to the abundant distribution of CuS/Cu<sub>2</sub>S–CdS boundaries and Cu<sub>2</sub>S sites within the sulfide-derived-CuCd<sub>2</sub> catalyst identified through electrochemical, spectroscopic, and

microscopic analyses. The electrochemically stable CdS phase plays a crucial role in creating the favorable catalyst structure, which stimulated C–C coupling and further reduction to ethanol. Li *et al.*<sup>78</sup> investigated an electroreductive CO<sub>2</sub>-to-ethanol conversion process, catalyzed by Cu nanosheets that are uniformly decorated with small Cu<sub>2</sub>S nanocrystals. The system, housed in an H-cell containing 0.1 M KHCO<sub>3</sub>, achieved a 20-hour stable high ethanol current density of approximately 20.7 mA cm<sup>-2</sup> with a large FE of 46% (at 1.2 V vs. RHE). Nanocrystal-decorated Cu nanosheets offer several advantages over heteroatom doping, Cu sulfide/oxide/nitride, and the parent catalyst, including: (i) a relatively high positive local charge of Cu<sup>δ+</sup>; (ii) effective interfaces between Cu<sup>δ+</sup> and Cu<sup>0</sup>; and (iii) a non-flat, stepped surface. These features improve CO<sub>2</sub> adsorption, enhance eCO<sub>2</sub>RR kinetics, facilitate the dissociation of the \*COOH intermediate into \*CO, strengthen the affinity of \*CO on the catalyst, and bring the energy barrier for C–C coupling below the desorption energy of \*CO. Liu *et al.*<sup>77</sup> found that incorporating Sm<sub>2</sub>O<sub>3</sub> into an S-doped Cu matrix creates a mixed phase of heterogeneous Sm and homogeneous S co-doped Cu, which stabilizes Cu<sup>δ+</sup> species and encourages the HCOOH formation pathway. With the



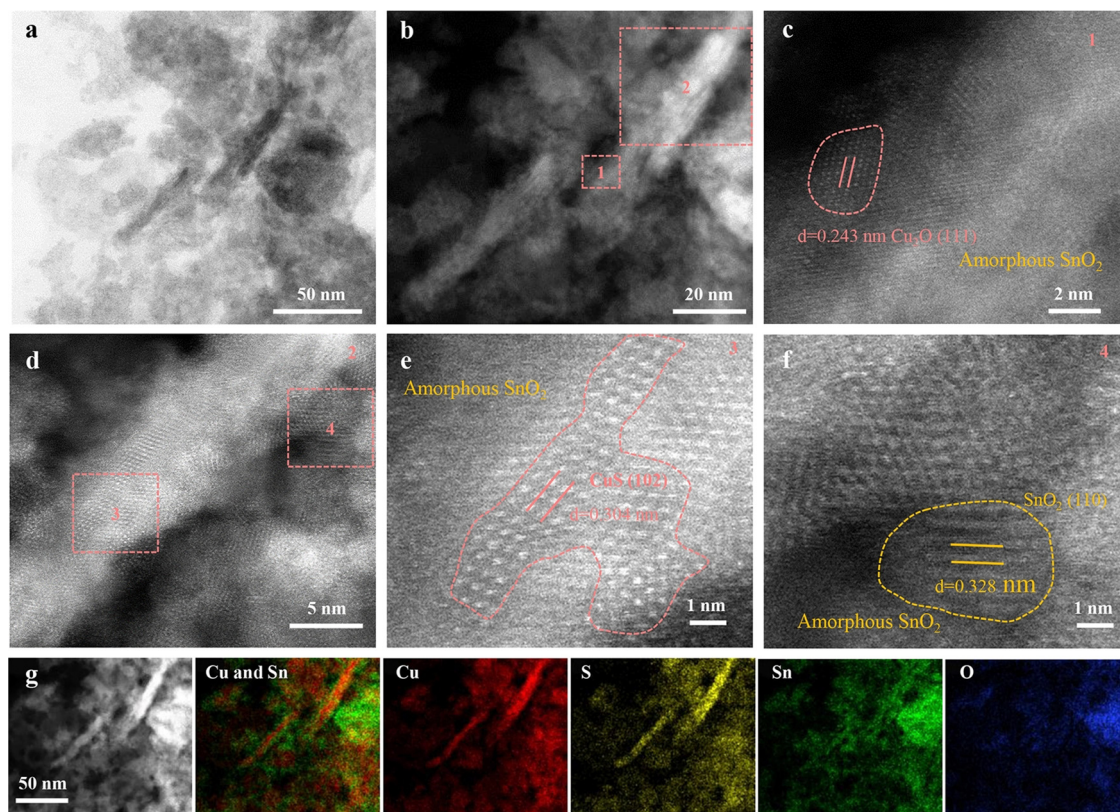


Fig. 7 Characterization of  $\text{Cu}_2\text{SnS}_3$  after  $\text{CO}_2\text{RR}$  (R- $\text{Cu}_2\text{SnS}_3$ ). (a) TEM image. (b)–(f) HAAD-STEM images. (g) HAAD-STEM image and EDX element mapping. Copyright 2021 Wiley-VCH.<sup>73</sup>

$\text{Sm}_{0.06}\text{-CuS}_{0.03}$  catalyst, they achieved a 92.1% FE for  $\text{HCOOH}$  production for 24 hours and an impressive current density of  $300 \text{ mA cm}^{-2}$  at a low reduction potential of  $-0.52 \text{ V vs. RHE}$ . This good electrocatalytic performance could, based on material characterizations and *in situ* experiments, be attributed to synergistic interaction between heterogeneously doped Sm species and homogeneously doped S elements. To explore the synergy between Sm and S, homogeneous S doping in the Cu active phase has been studied. This modification alters the inherent electronic structure of Cu, generating a surplus of  $\text{Cu}^{\delta+}$  ( $\delta = +0.61$ ). The presence of heterogeneous Sm species not only stabilizes  $\text{Cu}^{\delta+}$  sites but also raises the concentration of H ions, which enhances the adsorption of oxygen-containing intermediates ( $^*\text{OCHO}$ ) during  $\text{HCOOH}$  formation.

### 3.3 Size and morphology engineering

This section comments on recent approaches to customize  $\text{CuS}^{60,62,67,70,80,82,96}$  catalyst properties by particle size and morphology engineering. Through the optimization of these aspects, researchers continue to unveil new opportunities of copper sulfides in electrochemical  $\text{CO}_2$  reduction.

Shinagawa *et al.*<sup>67</sup> identified a relationship between the particle size and the formate selectivity in sulfur-modified copper catalysts. The particle size varies between fresh and used samples. After the electrochemical test, the initial average particle size of about  $15 \pm 7 \text{ nm}$  for a particular CuS samples reduced to approximately 3 nm, while that of another sample

decreased from roughly 42 nm to around 5 nm. They found that CuS-4 with larger particle sizes and higher sulfur contents resulted in increased FE for formate. Submicron-sized CuS catalysts, prepared solvothermally, outperformed their nano-sized counterparts. In regard to stability, we note that the Cu-S catalyst sustained its exclusive formate production over 12 hours at  $-0.8 \text{ V vs. RHE}$ . He *et al.*<sup>96</sup> uncovered a relationship between electrochemical parameters like the applied potential and charge passed and the extent of electrochemically-driven cation exchange reactions through structural and compositional analyses. This allowed them to systematically adjust binary cation ratios in nanocrystalline films, while preserving their initial 3D structural characteristics and engineer highly reactive catalytic sites by generating ultra-small structures and controlling surface oxidation states. Using this approach, they synthesized a  $\text{Cu}_2\text{S}$  electrocatalyst with high selectivity and activity for  $\text{eCO}_2\text{RR}$ , achieving a  $\text{CO}_2$  partial current density over  $19 \text{ mA cm}^{-2}$  for formate production at about  $-0.9 \text{ V vs. RHE}$  with a FE of 87.3% (stable for 9 hours at  $-0.82 \text{ V vs. NHE}$ ). This performance exceeds that of all Cu-based catalysts previously reported, providing insights into targeting specific reactive sites within nanocrystalline metal sulfides for  $\text{eCO}_2\text{RR}$  and suggesting a method for designing advanced electrocatalysts for various energy conversion processes. In their research, Lim *et al.*<sup>62</sup> developed highly selective and stable copper sulfide ( $\text{CuS}_x$ ) catalysts capable of converting industrial  $\text{CO}_2$  into formate. The copper foil spontaneously reacted with sulfur

species, increasing the average size and surface density of the  $\text{CuS}_x$  nanoparticles as the  $\text{H}_2\text{S}$  concentration rose, resulting in an increase of the formate FE from 22.7% to 71.8%.  $\text{CuS}_x$  catalysts demonstrated efficacy in both industrial and pure  $\text{CO}_2$  saturated electrolytes and maintained stability for 12 hours of continuous operation. The high FE for formate achieved by the  $\text{CuS}_x$  catalysts was due to the significant sulfur content, enhanced size and augmented surface density of  $\text{CuS}_x$  nanoparticles. Gao *et al.*<sup>80</sup> investigated the synthesis of  $\text{CuS}$  catalysts using various sulfur precursors for electrochemical  $\text{CO}_2$  reduction, and found that these precursors influence the catalyst performance due to different morphological features and  $\text{CO}$  selectivity. The  $\text{CuS}$  thiourea catalyst, with its distinctive morphology resembling nanoflowers, outperformed others by achieving the highest  $\text{CO}$  FE of 72.67% and favorable  $\text{CO}_2$  reduction kinetics.

Generally, the strategic engineering of catalysts' size and morphology selectively enhances  $\text{CO}_2$  reduction, while minimizing HER. This approach entails increasing exposure of boundary density and active sites, controlling exposed catalyst facets, and adjusting particle sizes. Furthermore, a refined morphology facilitates better mass transfer and encourages favorable kinetics for  $\text{CO}_2$  reduction reactions.

## 4. Recent strategies to enhance the performance of copper selenide in $\text{eCO}_2\text{RR}$

Similar as for  $\text{CuS}$ , recent strategies to improve the performance of copper selenide in the electrochemical  $\text{CO}_2$  reduction include structural and compositional tuning,<sup>64,83–85,87,88,103,104</sup> size and morphology engineering,<sup>83,89,92</sup> and heterostructure design.<sup>86</sup> Table 2 provides a compact summary of the latest results obtained on  $\text{CuSe}$ -based  $\text{eCO}_2\text{RR}$  catalysts, detailing their composition, preparation method, performance and product selectivity according to the electrolyte used. A comparison

of the  $\text{CO}_2$  electroreduction efficiency of copper chalcogenides prepared by alternative methods to those employed for the current leading catalysts is depicted in Fig. 9.

The most common strategy for designing better catalysts is to optimize the catalyst's structure and composition. Wang *et al.*<sup>85</sup> designed bimetallic  $\text{CuInSe}_2$  with Se vacancies to mimic the  $\text{eCO}_2\text{RR}$  catalytic properties of Au, resulting in an efficient conversion to  $\text{CO}$  accompanied by a suppressed HER activity. The orbital interaction of both metals with Se vacancies played a critical role in the  $\text{eCO}_2\text{RR}$  pathway. Se-defective  $\text{CuInSe}_2$  ( $\text{V-CuInSe}_2$ ) nanosheets exhibited an effective performance, illustrating the potential of cost-effective, noble-metal-like catalysts with unique electronic, structural, and catalytic properties. Duan *et al.*<sup>88</sup> suggested using natural superlattices with alternating active/conductive layers to stabilize the metal oxidation states, improving the  $\text{eCO}_2\text{RR}$  performance. They utilized  $\text{BiCuSeO}$  that features alternating conductive  $[\text{Cu}_2\text{Se}_2]^{2-}$  and insulating  $[\text{Bi}_2\text{O}_2]^{2+}$  sublayers as a representative example. It was demonstrated that the bismuth oxide layers played a significant role in activating  $\text{CO}_2$ , with created electrons being transferred quickly through the  $[\text{Cu}_2\text{Se}_2]^{2-}$  sublayers. DFT computations suggested that the Bi–O coordination in  $[\text{Bi}_2\text{O}_2]^{2+}$  effectively activates and stabilizes  $\text{OCHO}^*$  intermediates, which facilitates the selective  $\text{CO}_2$  electroreduction to formate. As a result, the naturally occurring  $\text{BiCuSeO}$  superlattices generate formate with a FE over 90% across a broad potential range between  $-0.4$  and  $-1.1$  V in a neutral electrolyte. This work demonstrates the use of natural superlattices for enhanced  $\text{eCO}_2\text{RR}$  selectivity and highlights the importance of specific coordination structures (see design strategies in Fig. 8). Ding *et al.*<sup>87</sup> found that exposed (220) crystal planes facilitate the adsorption of a key bridge-bonded  $^*\text{CO}_\text{B}$  intermediate. Potassium doping enhances the adsorption of linear  $^*\text{CO}_\text{L}$  and bridge  $^*\text{CO}_\text{B}$  intermediates on the catalyst surface, promoting C–C coupling to produce ethanol by protecting  $\text{Cu}(\text{I})$  sites during the electrocatalysis. The optimal K11.2%- $\text{Cu}_2\text{Se}$  nanosheet array achieves a 70.3% FE and a partial current density of  $35.8 \text{ mA cm}^{-2}$  for ethanol at  $-0.8$  V in  $0.1 \text{ M KHCO}_3$ , while remaining stable for

**Table 2** Performance summary of recently reported copper selenide  $\text{eCO}_2\text{RR}$  catalysts where partial current densities were not provided in the literature they were calculated by multiplying the total current density with the Faradaic efficiency of the main product

Preparation method	Catalysts	Electrolyte	$E$ (V vs. RHE)	Partial current density ( $\text{mA cm}^{-2}$ )	Product and FE	Ref.
Solvothermal	$\text{BiCuSeO}$ superlattices	$0.5 \text{ M KHCO}_3$	$-0.9$	18.7	$\text{HCOO}^-$ ( $\sim 93.4\%$ )	88
		$[\text{BMIM}]\text{PF}_6$ (30 wt%)/ $\text{CH}_3\text{CN}/\text{H}_2\text{O}$ (5 wt%)	$-2.1 \text{ V vs. Ag/Ag}^+$	32.2	$\text{CH}_3\text{OH}$ (77.6%)	83
	Se-defective $\text{CuInSe}_2$	$0.5 \text{ M KHCO}_3$	$-0.7$	112	$\text{CO}$ (91%)	85
	Nanostructured $\text{Cu}_2\text{Se}$	$0.3 \text{ M NaHCO}_3$	$-1.3$	NA	$\text{HCOO}^-$ (94.2%)	84
	2D $\text{CuSe/g-C}_3\text{N}_4$	$0.1 \text{ M KHCO}_3$	$-1.2$	$\sim 8.4$	$\text{CO}$ (85.28%)	86
	K11.2%- $\text{Cu}_2\text{Se}$ nanosheets	$0.1 \text{ M KHCO}_3$	$-0.8$	35.8	$\text{C}_2\text{H}_5\text{OH}$ (70.3%)	87
	$\text{Cu}_{1.22}\text{V}_{0.19}\text{Se}$ nanotubes	$0.1 \text{ M KHCO}_3$	$-0.8$	207.9	$\text{C}_2\text{H}_5\text{OH}$ (68.3%)	105
	$\text{Cu}_3\text{Se}_2$ nanosheet on Cu foam	$0.1 \text{ M KHCO}_3$	$-1.2$	11.9	$\text{CO}$ ( $> 40\%$ )	89
	2D $\text{Cu}_{2-x}\text{Se}$ with numerous Se vacancies	$0.5 \text{ M KHCO}_3$	$-0.8$	7.44	$\text{C}_2\text{H}_5\text{OH}$ (68.1%)	106
	$\text{Cu}_{1.82}\text{Se}$ nanowires	$0.1 \text{ M KHCO}_3$	$-1.1$	10.2	$\text{C}_2\text{H}_4$ (55%)	64
Wet chemistry and annealing	N-CuSe	$0.5 \text{ M KHCO}_3$	$-1.1$	25.8	$\text{HCOO}^-$ (55.2%)	104
Electro-synthesis	3D $\text{CuSe}$ nanocubes	$1.0 \text{ M KHCO}_3$ or	$-1.3$	10	NA	92
		$0.1 \text{ M } [\text{BMIM}]\text{PF}_6$				

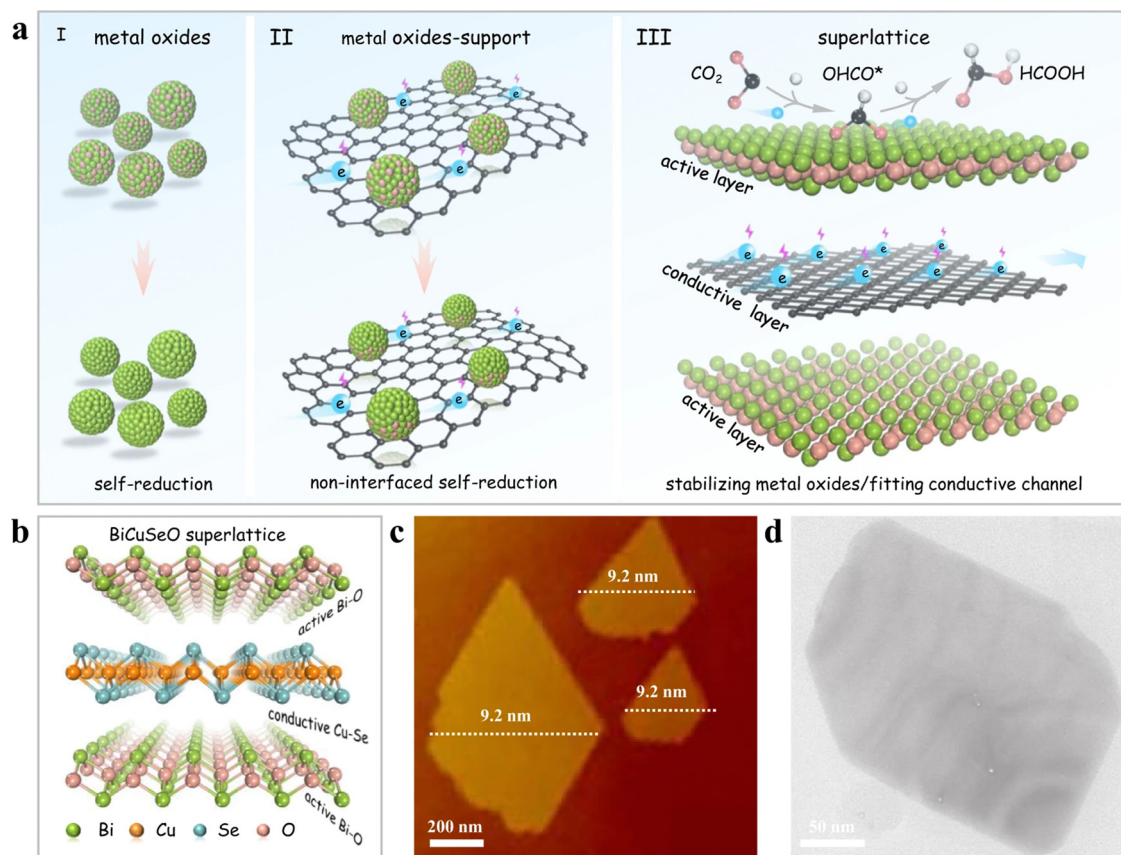


Fig. 8 Development of BiCuSeO superlattice nanosheets: design strategy and structural analysis. (a) Visual representation of single-phase materials using alternating active and conductive layer arrangements, including (I) pure metal oxides, (II) metal oxides interacting with a support, and (III) a superlattice made up of vertically stacked active (metal oxides) and conductive CuSe layers in a one-to-one ratio. (b) Structural representation of the BiCuSeO superlattice, (c) AFM image, and (d) TEM image. Copyright 2022 Nature Publishing Group.<sup>88</sup>

130 hours. With a FE of ethanol exceeding 50% in a wide potential range from  $-0.6$  to  $-1.2$  V, this approach demonstrates the good potential of these superlattices for real-world implementations. This study presents a novel method to enhance  $\text{CO}_2\text{RR}$  to ethanol through alkali metal doping and offers a rational strategy for designing catalysts with high product selectivity for multiple proton-coupled electron transfer reactions. Sun *et al.*<sup>105</sup> developed V-doped  $\text{Cu}_2\text{Se}$  nanotubes for flow-cell  $\text{CO}_2\text{RR}$ , efficiently producing ethanol. The introduction of  $\text{V}^{4+}$  ions into  $\text{Cu}_2\text{Se}$  provides diverse active sites, prevents  $\text{Cu}^+$  reduction, and facilitates different CO intermediates' adsorption, leading to ethanol formation, as confirmed by *in situ* DRIFTS and DFT calculations. The optimized  $\text{Cu}_{1.22}\text{V}_{0.19}\text{Se}$  nanotubes achieve a 68.3%  $\text{FE}_{\text{C}_2\text{H}_5\text{OH}}$  and a  $-207.9 \text{ mA cm}^{-2}$  partial current density for ethanol at  $-0.8$  V in 1 M KOH. Hao *et al.*<sup>89</sup> presented self-supported  $\text{Cu}_3\text{Se}_2$  featuring a hierarchical structure composed of nanosheet-assembled fibers whose properties are modulated by anions. This structure delivers a flexible and broad range of  $\text{H}_2$ -to-CO ratios (from 0.8 to 6.0) within an extensive potential window and boasts an impressive turnover frequency (TOF) of  $1303 \text{ h}^{-1}$ . Wang *et al.*<sup>106</sup> prepared 2D  $\text{Cu}_{2-x}\text{Se}$  with numerous Se vacancies, enhancing  $\text{CO}_2$  to ethanol conversion activity and selectivity during  $\text{eCO}_2\text{RR}$ . The optimal Cu-Cu spacing in this ultra-thin

material, obtained by the presence of Se vacancies, promotes C-C coupling, resulting in efficient ethanol production. Specifically,  $\text{V}_{\text{Se}}\text{-Cu}_{2-x}\text{Se}$ , with a  $2.51 \text{ \AA}$  Cu-Cu distance, achieves a 68.1% faradaic efficiency and  $7.44 \text{ mA cm}^{-2}$  partial current density for ethanol in 0.5 M  $\text{KHCO}_3$ , within a  $-0.4$  to  $-1.6$  V potential range. *In situ* Fourier transform infrared spectroscopy and DFT calculations confirm that this spacing optimizes the Gibbs free energy for both ethanol generation and formate formation steps.

Yang *et al.*<sup>83</sup> emphasized that the copper selenide catalysts' performance was strongly influenced by their size and morphology, which could be fine-tuned using the volume ratio of diethylenetriamine and water (VDETA/ $\text{VH}_2\text{O}$ ). They found that the catalysts synthesized at a VDETA/ $\text{VH}_2\text{O}$  ratio of 1/3 had the smallest size, resulting in the highest current density and FE. The  $\text{Cu}_{1.63}\text{Se}(1/3)$  nanocatalysts achieved an exceptional current density of  $41.5 \text{ mA cm}^{-2}$  and an FE of 77.6% at a potential of  $-2.1$  V versus  $\text{Ag}/\text{Ag}^+$ .

Since the first application of copper selenide in the electrocatalytic carbon dioxide reduction in 2019,<sup>83</sup> the field has seen substantial progress. This advancement is highlighted by the diversification of the reaction products that now include methanol, formic acid, carbon monoxide (CO), and notably, ethanol.<sup>87,105,106</sup> Additionally, there has been a remarkable



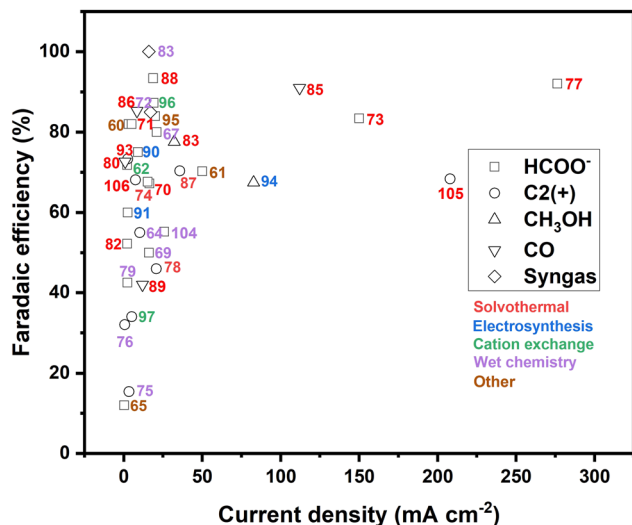


Fig. 9 Comparison of the CO<sub>2</sub> electroreduction performance of copper chalcogenides in different reported state-of-art catalysts. The numbers in the figure are the corresponding reference numbers. The symbols reflect the main eCO<sub>2</sub>RR product of each catalyst, and the color its fabrication method.

increase in electrical current efficiency, rising from 41 mA cm<sup>-2</sup> to 207.9 mA cm<sup>-2</sup>. These improvements that are largely due to the implementation of new defect engineering and doping strategies will drive further research interest in this area.

## 5. Challenges and future directions

Among the family of copper chalcogenides, copper sulfide and copper selenide have been extensively utilized for electrochemical CO<sub>2</sub> reduction while copper telluride has only been employed for water splitting applications.<sup>107,108</sup> Copper telluride, due to its distinct electronic structure and stability compared to its predecessors, may bring more possibilities for future eCO<sub>2</sub>RR research.

Despite significant progress in the application of copper chalcogenide materials as efficient electro-catalysts for eCO<sub>2</sub>RR, moving from products like CO, formate to methanol and C2+ products, and the transition from low to high current densities up to 300 mA cm<sup>-2</sup>, numerous challenges remain to be tackled. These include the use of more rational design strategies for increasing selectivity towards specific products and enhancing the catalysts' long-term stability under operational conditions. Understanding the relationship between catalyst structure, product selectivity, and long-term stability is crucial. *In situ* and *operando* characterization techniques can contribute to the understanding of structure-performance relationships and the identification of active sites in these materials.

As the synthesis of catalysts typically does not involve a unique synthetic method but multiple ones, it is vital to explore and optimize synergistic effects that can be obtained by integration of various synthetic strategies. This can result in the advancement of novel copper chalcogenide materials with unique properties and improved eCO<sub>2</sub>RR performance.

A major challenge in eCO<sub>2</sub>RR using copper chalcogenides is achieving a high selectivity towards specific products such as formate, CO or hydrocarbons. Subsequent investigations should prioritize the development of strategies for tailoring the electronic properties of copper chalcogenide materials using elemental doping and adjusting their geometric properties through specific chemical synthesis methods, as these can directly influence their catalytic activity and selectivity.

Another important point is the stability of copper chalcogenide catalysts during long-term eCO<sub>2</sub>RR operation, which is crucial for practical applications. Degradation, agglomeration, and morphological changes can lead to performance deterioration over time. Addressing these stability issues requires the development of novel catalyst architectures and protective strategies, such as encapsulating the catalysts in robust matrices or engineering support materials to prevent degradation.

Optimizing the performance of copper chalcogenide materials requires developing rational design strategies that can be achieved by integrating advanced characterization techniques, including *in situ* and *operando* methods, with computational modeling to acquire understanding of active sites and reaction mechanisms. Such an approach will enable the strategic development of advanced copper chalcogenide materials with enhanced eCO<sub>2</sub>RR performance.

For practical applications of copper chalcogenide catalysts in large-scale eCO<sub>2</sub>RR, it is essential to address the challenges associated with their scalability and environmental impact. Future research should focus on developing cost-effective and environmentally friendly synthetic methods, as well as evaluating the life cycle and techno-economic aspects of these materials.

In conclusion, overcoming the challenges in enhancing the selectivity and stability of copper chalcogenide materials will require a combination of rational design strategies, advanced characterization techniques, and the integration of various synthetic methods. Addressing these challenges is crucial for realizing the full potential of copper chalcogenides in CO<sub>2</sub> electroreduction and contributing towards a sustainable future.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This project has received funding from the European Union's Horizon 2020 MSCA-ITN programme under grant agreement No 955650 (CATCHY).

## References

- 1 S. Nitopi, E. Bertheussen, S. B. Scott, X. Liu, A. K. Engstfeld, S. Horch, B. Seger, I. E. L. Stephens, K. Chan, C. Hahn, J. K. Nørskov, T. F. Jaramillo and I. Chorkendorff, *Chem. Rev.*, 2019, **119**, 7610–7672.



- 2 T. K. Todorova, M. W. Schreiber and M. Fontecave, *ACS Catal.*, 2019, **10**, 1754–1768.
- 3 G. M. Tomboc, S. Choi, T. Kwon, Y. J. Hwang and K. Lee, *Adv. Mater.*, 2020, **32**, e1908398.
- 4 L. Zhang, I. Merino-Garcia, J. Albo and C. M. Sánchez-Sánchez, *Curr. Opin. Electrochem.*, 2020, **23**, 65–73.
- 5 J. Zhao, S. Xue, J. Barber, Y. Zhou, J. Meng and X. Ke, *J. Mater. Chem. A*, 2020, **8**, 4700–4734.
- 6 S. Bhardwaj, A. Biswas, M. Das and R. S. Dey, *Sustain. Energy Fuels*, 2021, **5**, 2393–2414.
- 7 Y. Jia, F. Li, K. Fan and L. Sun, *Adv. Powder Mater.*, 2022, **1**, 100012.
- 8 D. Karapinar, C. E. Creissen, J. G. Rivera de la Cruz, M. W. Schreiber and M. Fontecave, *ACS Energy Lett.*, 2021, **6**, 694–706.
- 9 K. Van Daele, B. De Mot, M. Pupo, N. Daems, D. Pant, R. Kortlever and T. Breugelmans, *ACS Energy Lett.*, 2021, **6**, 4317–4327.
- 10 O. Gutiérrez-Sánchez, B. de Mot, N. Daems, M. Bulut, J. Vaes, D. Pant and T. Breugelmans, *Energy Fuels*, 2022, **36**, 13115–13123.
- 11 J. Mertens, C. Breyer, K. Arning, A. Bardow, R. Belmans, A. Dibenedetto, S. Erkman, J. Griepkoven, G. Léonard, S. Nizou, D. Pant, A. S. Reis-Machado, P. Styring, J. Vente, M. Webber and C. J. Sapart, *Joule*, 2023, **7**, 442–449.
- 12 C. Liu, J. Gong, Z. Gao, L. Xiao, G. Wang, J. Lu and L. Zhuang, *Sci. China: Chem.*, 2021, **64**, 1660–1678.
- 13 W. Ma, X. He, W. Wang, S. Xie, Q. Zhang and Y. Wang, *Chem. Soc. Rev.*, 2021, **50**, 12897–12914.
- 14 W. Quan, Y. Lin, Y. Luo and Y. Huang, *Adv. Sci.*, 2021, **8**, 2101597.
- 15 Z. Sun, Y. Hu, D. Zhou, M. Sun, S. Wang and W. Chen, *ACS Energy Lett.*, 2021, **6**, 3992–4022.
- 16 B. Talukdar, S. Mendiratta, M. H. Huang and C. H. Kuo, *Chem. – Asian J.*, 2021, **16**, 2168–2184.
- 17 C. Xiao and J. Zhang, *ACS Nano*, 2021, **15**, 7975–8000.
- 18 Y. Xin, K. Yu, L. Zhang, Y. Yang, H. Yuan, H. Li, L. Wang and J. Zeng, *Adv. Mater.*, 2021, **33**, e2008145.
- 19 B. Deng, X. Zhao, Y. Li, M. Huang, S. Zhang and F. Dong, *Sci. China: Chem.*, 2022, **66**, 78–95.
- 20 H. Li, P. Wei, D. Gao and G. Wang, *Curr. Opin. Green Sustain. Chem.*, 2022, **34**, 100589.
- 21 S. Mohan, B. Honnappa, A. Augustin, M. Shanmugam, C. Chuaicham, K. Sasaki, B. Ramasamy and K. Sekar, *Catalysts*, 2022, **12**, 445.
- 22 S. Mu, H. Lu, Q. Wu, L. Li, R. Zhao, C. Long and C. Cui, *Nat. Commun.*, 2022, **13**, 3694.
- 23 L. Zaza, K. Rossi and R. Buonsanti, *ACS Energy Lett.*, 2022, **7**, 1284–1291.
- 24 Z. Zhang, L. Bian, H. Tian, Y. Liu, Y. Bando, Y. Yamauchi and Z. L. Wang, *Small*, 2022, **18**, 2107450.
- 25 C. Zhu, S. Zhao, G. Shi and L. Zhang, *ChemSusChem*, 2022, **15**, e202200068.
- 26 Y. Lei, Z. Wang, A. Bao, X. Tang, X. Huang, H. Yi, S. Zhao, T. Sun, J. Wang and F. Gao, *J. Chem. Eng.*, 2023, **453**, 139663.
- 27 Y. Yang, S. Louisia, S. Yu, J. Jin, I. Roh, C. Chen, M. V. Fonseca Guzman, J. Feijoo, P. C. Chen, H. Wang, C. J. Pollock, X. Huang, Y. T. Shao, C. Wang, D. A. Muller, H. D. Abruna and P. Yang, *Nature*, 2023, **614**, 262–269.
- 28 C. Choi, S. Kwon, T. Cheng, M. Xu, P. Tieu, C. Lee, J. Cai, H. M. Lee, X. Pan, X. Duan, W. A. Goddard and Y. Huang, *Nat. Catal.*, 2020, **3**, 804–812.
- 29 J. Bu, Z. Liu, W. Ma, L. Zhang, T. Wang, H. Zhang, Q. Zhang, X. Feng and J. Zhang, *Nat. Catal.*, 2021, **4**, 557–564.
- 30 C. Durante, *Nat. Catal.*, 2021, **4**, 537–538.
- 31 C. Kim, J. C. Bui, X. Luo, J. K. Cooper, A. Kusoglu, A. Z. Weber and A. T. Bell, *Nat. Energy*, 2021, **6**, 1026–1034.
- 32 F. Li, X. V. Medvedeva, J. J. Medvedev, E. Khairullina, H. Engelhardt, S. Chandrasekar, Y. Guo, J. Jin, A. Lee, H. Thérien-Aubin, A. Ahmed, Y. Pang and A. Klinkova, *Nat. Catal.*, 2021, **4**, 479–487.
- 33 M. C. O. Monteiro, F. Dattila, B. Hagedoorn, R. García-Muelas, N. López and M. T. M. Koper, *Nat. Catal.*, 2021, **4**, 654–662.
- 34 R. Shi, Z. Wang, Y. Zhao, G. I. N. Waterhouse, Z. Li, B. Zhang, Z. Sun, C. Xia, H. Wang and T. Zhang, *Nat. Catal.*, 2021, **4**, 565–574.
- 35 H. Shin, K. U. Hansen and F. Jiao, *Nat. Sustain.*, 2021, **4**, 911–919.
- 36 Z. Xing, L. Hu, D. S. Ripatti, X. Hu and X. Feng, *Nat. Commun.*, 2021, **12**, 136.
- 37 T. Zheng, C. Liu, C. Guo, M. Zhang, X. Li, Q. Jiang, W. Xue, H. Li, A. Li, C. W. Pao, J. Xiao, C. Xia and J. Zeng, *Nat. Nanotechnol.*, 2021, **16**, 1386–1393.
- 38 Z. Han, D. Han, Z. Chen, J. Gao, G. Jiang, X. Wang, S. Lyu, Y. Guo, C. Geng, L. Yin, Z. Weng and Q. H. Yang, *Nat. Commun.*, 2022, **13**, 3158.
- 39 J. Timoshenko, A. Bergmann, C. Rettenmaier, A. Herzog, R. M. Arán-Ais, H. S. Jeon, F. T. Haase, U. Hejral, P. Grosse, S. Köhl, E. M. Davis, J. Tian, O. Magnussen and B. Roldan Cuenya, *Nat. Catal.*, 2022, **5**, 259–267.
- 40 X. Wang, P. Ou, A. Ozden, S.-F. Hung, J. Tam, C. M. Gabardo, J. Y. Howe, J. Sisler, K. Bertens, F. P. García de Arquer, R. K. Miao, C. P. O'Brien, Z. Wang, J. Abed, A. S. Rasouli, M. Sun, A. H. Ip, D. Sinton and E. H. Sargent, *Nat. Energy*, 2022, **7**, 170–176.
- 41 Y. Xie, P. Ou, X. Wang, Z. Xu, Y. C. Li, Z. Wang, J. E. Huang, J. Wicks, C. McCallum, N. Wang, Y. Wang, T. Chen, B. T. W. Lo, D. Sinton, J. C. Yu, Y. Wang and E. H. Sargent, *Nat. Catal.*, 2022, **5**, 564–570.
- 42 A. Xu, S.-F. Hung, A. Cao, Z. Wang, N. Karmodak, J. E. Huang, Y. Yan, A. Sedighian Rasouli, A. Ozden, F.-Y. Wu, Z.-Y. Lin, H.-J. Tsai, T.-J. Lee, F. Li, M. Luo, Y. Wang, X. Wang, J. Abed, Z. Wang, D.-H. Nam, Y. C. Li, A. H. Ip, D. Sinton, C. Dong and E. H. Sargent, *Nat. Catal.*, 2022, **5**, 1081–1088.
- 43 T. Zhang, J. C. Bui, Z. Li, A. T. Bell, A. Z. Weber and J. Wu, *Nat. Catal.*, 2022, **5**, 202–211.
- 44 B. Seger, M. Robert and F. Jiao, *Nat. Sustain.*, 2023, **6**, 236–238.

- 45 D. R. Brown, T. Day, K. A. Borup, S. Christensen, B. B. Iversen and G. J. Snyder, *APL Mater.*, 2013, **1**, 052107.
- 46 C. Han, Q. Sun, Z. Li and S. X. Dou, *Adv. Energy Mater.*, 2016, **6**, 1600498.
- 47 Y. Shi, C. Sturm and H. Kleinke, *J. Solid State Chem.*, 2019, **270**, 273–279.
- 48 T. Deng, T. Xing, M. K. Brod, Y. Sheng, P. Qiu, I. Veremchuk, Q. Song, T.-R. Wei, J. Yang, G. J. Snyder, Y. Grin, L. Chen and X. Shi, *Energy Environ. Sci.*, 2020, **13**, 3041–3053.
- 49 J. Jiang, H. Zhu, Y. Niu, Q. Zhu, S. Song, T. Zhou, C. Wang and Z. Ren, *J. Mater. Chem. A*, 2020, **8**, 4790–4799.
- 50 M. M. Kubenova, K. A. Kuterbekov, M. K. Balapanov, R. K. Ishembetov, A. M. Kabyshev and K. Z. Bekmyrza, *Nanomater.*, 2021, **11**, 2238.
- 51 J. R.-F. Ilka Kriegel, A. Wisnet, H. Zhang, C. Waurisch, A. Eychmüller, A. Dubavik, A. O. Govorov and J. Feldmann, *ACS Nano*, 2013, **7**, 4367–4377.
- 52 B. Hamawandi, S. Ballikaya, M. Rasander, J. Halim, L. Vinciguerra, J. Rosen, M. Johnsson and M. Toprak, *Nanomater.*, 2020, **10**, 854.
- 53 T. Willhammar, K. Sentosun, S. Mourdikoudis, B. Goris, M. Kurttepel, M. Bercx, D. Lamoën, B. Partoens, I. Pastoriza-Santos, J. Perez-Juste, L. M. Liz-Marzan, S. Bals and G. Van Tendeloo, *Nat. Commun.*, 2017, **8**, 14925.
- 54 X. Chen, J. Yang, T. Wu, L. Li, W. Luo, W. Jiang and L. Wang, *Nanoscale*, 2018, **10**, 15130–15163.
- 55 W. Li, A. Shavel, R. Guzman, J. Rubio-Garcia, C. Flox, J. Fan, D. Cadavid, M. Ibanez, J. Arbiol, J. R. Morante and A. Cabot, *Chem. Commun.*, 2011, **47**, 10332–10334.
- 56 Y. Zhao and C. Burda, *Energy Environ. Sci.*, 2012, **5**, 5564–5576.
- 57 C. Coughlan, M. Ibanez, O. Dobrozhan, A. Singh, A. Cabot and K. M. Ryan, *Chem. Rev.*, 2017, **117**, 5865–6109.
- 58 B. Yun, H. Zhu, J. Yuan, Q. Sun and Z. Li, *J. Mater. Chem. B*, 2020, **8**, 4778–4812.
- 59 Y. Chen, K. Chen, J. Fu, A. Yamaguchi, H. Li, H. Pan, J. Hu, M. Miyauchi and M. Liu, *Nano Mater. Sci.*, 2020, **2**, 235–247.
- 60 B. Zhang, M. Wang, J. Ding, Y. Li, G. Cao, M. T. Bernards, Y. He and Y. Shi, *J. CO<sub>2</sub> Util.*, 2020, **39**, 101169.
- 61 T. Dou, Y. Qin, F. Zhang and X. Lei, *ACS Appl. Energy Mater.*, 2021, **4**, 4376–4384.
- 62 J. W. Lim, W. J. Dong, J. Y. Park, D. M. Hong and J. L. Lee, *ACS Appl. Mater. Interfaces*, 2020, **12**, 22891–22900.
- 63 K. Sun, S. Song, A. Kumar, Z. Shang, X. Duan, S. Wang, S. Tian, J. Tang, D. Zhou and X. Sun, *ACS Appl. Nano Mater.*, 2021, **4**, 2357–2364.
- 64 Y. Mi, X. Peng, X. Liu and J. Luo, *ACS Appl. Energy Mater.*, 2018, **1**, 5119–5123.
- 65 C. H. M. van Oversteeg, M. Tapia Rosales, K. H. Helfferich, M. Ghiasi, J. D. Meeldijk, N. J. Firet, P. Ngene, C. de Mello Donegá and P. E. de Jongh, *Catal. Today*, 2021, **377**, 157–165.
- 66 J. Xu, W. Zhang, Z. Yang, S. Ding, C. Zeng, L. Chen, Q. Wang and S. Yang, *Adv. Funct. Mater.*, 2009, **19**, 1759–1766.
- 67 T. Shinagawa, G. O. Larrazábal, A. J. Martín, F. Krumeich and J. Pérez-Ramírez, *ACS Catal.*, 2018, **8**, 837–844.
- 68 F. Liu, R. Xu, L. Hong, Z. Yang, Y. Li, C. Wang, H. Jia and C. Yang, *ChemistrySelect*, 2020, **5**, 360–368.
- 69 X. Zhang, R. Sa, F. Zhou, Y. Rui, R. Liu, Z. Wen and R. Wang, *CCS Chem.*, 2021, **3**, 199–207.
- 70 H. Yuan, Z. Liu, S. Sang and X. Wang, *Appl. Surf. Sci.*, 2023, **613**, 156130.
- 71 Z. Wu, J. Yu, K. Wu, J. Song, H. Gao, H. Shen, X. Xia, W. Lei and Q. Hao, *Appl. Surf. Sci.*, 2022, **575**, 151796.
- 72 X. Wang, J. Lv, J. Zhang, X. L. Wang, C. Xue, G. Bian, D. Li, Y. Wang and T. Wu, *Nanoscale*, 2020, **12**, 772–784.
- 73 W. Wang, Z. Wang, R. Yang, J. Duan, Y. Liu, A. Nie, H. Li, B. Y. Xia and T. Zhai, *Angew. Chem., Int. Ed.*, 2021, **60**, 22940–22947.
- 74 S. Wang, T. Kou, J. B. Varley, S. A. Akhade, S. E. Weitzner, S. E. Baker, E. B. Duoss and Y. Li, *ACS Mater. Lett.*, 2020, **3**, 100–109.
- 75 C. Peng, G. Luo, J. Zhang, M. Chen, Z. Wang, T. K. Sham, L. Zhang, Y. Li and G. Zheng, *Nat. Commun.*, 2021, **12**, 1580.
- 76 V. S. S. Mosali, X. Zhang, Y. Liang, L. Li, G. Puxty, M. D. Horne, A. Brajter-Toth, A. M. Bond and J. Zhang, *ChemSusChem*, 2021, **14**, 2924–2934.
- 77 J. Liu, P. Li, J. Bi, Y. Wang, Q. Zhu, X. Sun, J. Zhang, Z. Liu and B. Han, *Chin. J. Chem.*, 2023, **41**, 1443–1449.
- 78 Y. Li, Y. Chen, T. Chen, G. Shi, L. Zhu, Y. Sun and M. Yu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 18857–18866.
- 79 U. Legrand, R. Boudreault and J. L. Meunier, *Electrochim. Acta*, 2019, **318**, 142–150.
- 80 Y. Gao, Y. Guo, Y. Zou, W. Liu, Y. Luo, B. Liu and C. Zhao, *ACS Appl. Energy Mater.*, 2023, **6**, 1340–1354.
- 81 H. Cheng, S. Liu, J. Zhang, T. Zhou, N. Zhang, X. S. Zheng, W. Chu, Z. Hu, C. Wu and Y. Xie, *Nano Lett.*, 2020, **20**, 6097–6103.
- 82 Y. T. J. Chen, Y. Zou, X. Li and J. Jiang, *Mater. Lett.*, 2021, **284**, 128919.
- 83 D. Yang, Q. Zhu, C. Chen, H. Liu, Z. Liu, Z. Zhao, X. Zhang, S. Liu and B. Han, *Nat. Commun.*, 2019, **10**, 677.
- 84 A. Saxena, W. Liyanage, J. Masud, S. Kapila and M. Nath, *J. Mater. Chem. A*, 2021, **9**, 7150–7161.
- 85 J. Wang, X. Zheng, G. Wang, Y. Cao, W. Ding, J. Zhang, H. Wu, J. Ding, H. Hu, X. Han, T. Ma, Y. Deng and W. Hu, *Adv. Mater.*, 2021, **34**, e2106354.
- 86 H. Zhang, T. Ouyang, J. Li, M. Mu and X. Yin, *Electrochim. Acta*, 2021, **390**, 138766.
- 87 L. Ding, N. Zhu, Y. Hu, Z. Chen, P. Song, T. Sheng, Z. Wu and Y. Xiong, *Angew. Chem., Int. Ed.*, 2022, **61**, e202209268.
- 88 J. Duan, T. Liu, Y. Zhao, R. Yang, Y. Zhao, W. Wang, Y. Liu, H. Li, Y. Li and T. Zhai, *Nat. Commun.*, 2022, **13**, 2039.
- 89 Y. Hao, Y. Sun, H. Wang, J. Xue, J. Ren, A. A. S. Devi, M. Y. Maximov, F. Hu and S. Peng, *Electrochim. Acta*, 2023, **449**, 142213.
- 90 Y. Deng, Y. Huang, D. Ren, A. D. Handoko, Z. W. Seh, P. Hirunsit and B. S. Yeo, *ACS Appl. Mater. Interfaces*, 2018, **10**, 28572–28581.

- 91 K. R. Phillips, Y. Katayama, J. Hwang and Y. Shao-Horn, *J. Phys. Chem. Lett.*, 2018, **9**, 4407–4412.
- 92 R. Elakkiya and G. Maduraiveeran, *Sustain. Energy Fuels.*, 2021, **5**, 6430–6440.
- 93 C. Guo, Y. Guo, Y. Shi, X. Lan, Y. Wang, Y. Yu and B. Zhang, *Angew. Chem., Int. Ed.*, 2022, **61**, e202205909.
- 94 P. Li, J. Bi, J. Liu, Q. Zhu, C. Chen, X. Sun, J. Zhang and B. Han, *Nat. Commun.*, 2022, **13**, 1965.
- 95 A. W. Kahsay, K. B. Ibrahim, M.-C. Tsai, M. K. Birhanu, S. A. Chala, W.-N. Su and B.-J. Hwang, *Catal. Lett.*, 2019, **149**, 860–869.
- 96 W. He, I. Liberman, I. Rozenberg, R. Ifraemov and I. Hod, *Angew. Chem., Int. Ed.*, 2020, **59**, 8262–8269.
- 97 J. Li, J. Li, C. Dun, W. Chen, D. Zhang, J. Gu, J. J. Urban and J. W. Ager, *RSC Adv.*, 2021, **11**, 23948–23959.
- 98 Z. Zhao and G. Lu, *Chem. Sci.*, 2022, **13**, 3880–3887.
- 99 Z. Zhao and G. Lu, *Adv. Energy Mater.*, 2023, **13**, 2203138.
- 100 M.-J. Cheng, E. L. Clark, H. H. Pham, A. T. Bell and M. Head-Gordon, *ACS Catal.*, 2016, **6**, 7769–7777.
- 101 M. Karamad, V. Tripkovic and J. Rossmeisl, *ACS Catal.*, 2014, **4**, 2268–2273.
- 102 F. Abild-Pedersen, J. Greeley, F. Studt, J. Rossmeisl, T. R. Muntter, P. G. Moses, E. Skulason, T. Bligaard and J. K. Nørskov, *Phys. Rev. Lett.*, 2007, **99**, 016105.
- 103 C. Brea and G. Hu, *J. Mater. Chem. A*, 2022, **10**, 10162–10170.
- 104 Y. Li, G. Shi, T. Chen, L. Zhu, D. Yu, Y. Sun, F. Besenbacher and M. Yu, *Appl. Catal., B*, 2022, **305**, 121080.
- 105 W. Sun, P. Wang, Y. Jiang, Z. Jiang, R. Long, Z. Chen, P. Song, T. Sheng, Z. Wu and Y. Xiong, *Adv. Mater.*, 2022, **34**, e2207691.
- 106 H. Wang, X. Bi, Y. Yan, Y. Zhao, Z. Yang, H. Ning and M. Wu, *Adv. Funct. Mater.*, 2023, **33**, 2214846.
- 107 X. Shao, X. Zhang, Y. Liu, J. Qiao, X.-D. Zhou, N. Xu, J. L. Malcombe, J. Yi and J. Zhang, *J. Mater. Chem. A*, 2021, **9**, 2526–2559.
- 108 S. Kumaravel, K. Karthick, P. Thiruvengadam, J. M. Johny, S. S. Sankar and S. Kundu, *Inorg. Chem.*, 2020, **59**, 11129–11141.